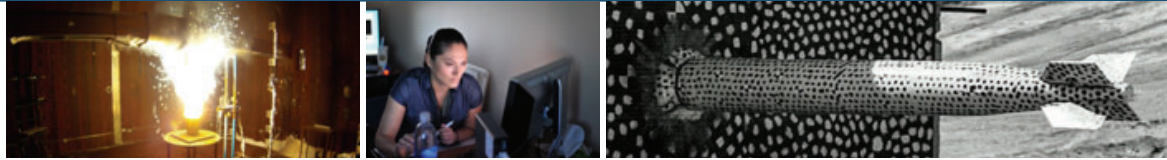




Sandia  
National  
Laboratories

# Multiscale Modeling of Materials for Fusion Energy



**Mary Alice Cusentino**

*Material, Physical, and Chemical Sciences Center  
Sandia National Laboratories*

ANS Lightning Talk  
February 23, 2021

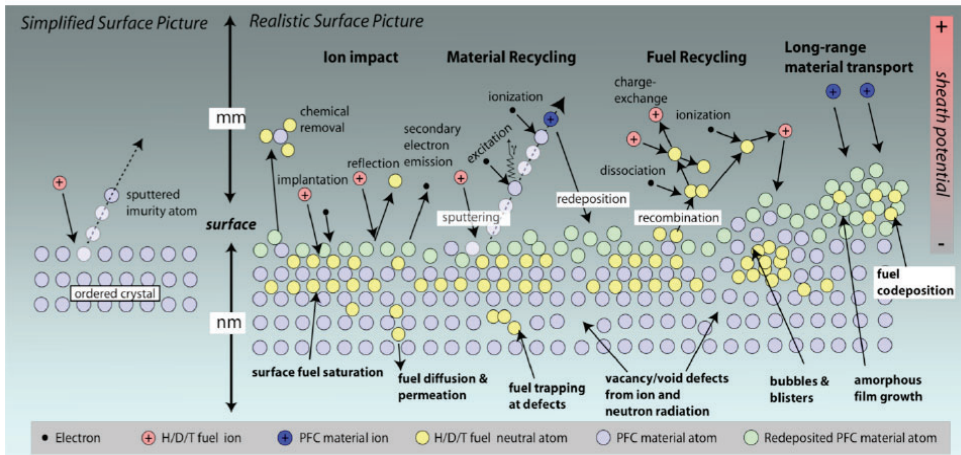
Approved for  
Release:  
SAND2021-1733 PE



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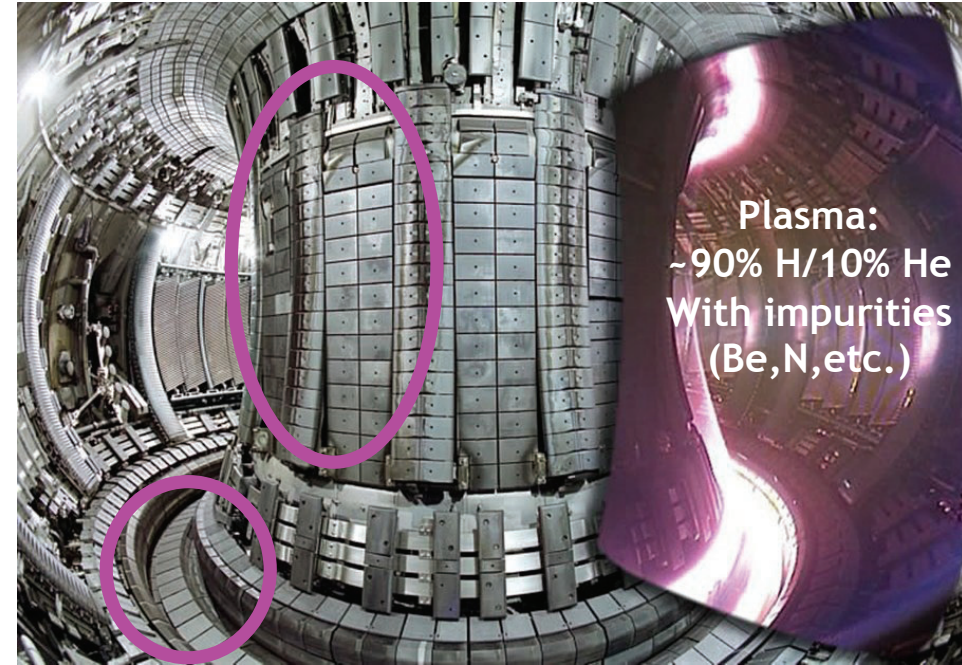
## 2 | Materials for Fusion Energy

- Difficult to develop materials to handle extreme conditions within tokamak
- Large heat loads of 10-20 MW/m<sup>3</sup>
- High particles fluxes of  $\sim 10^{24}$  m<sup>-2</sup>s<sup>-1</sup> of mixed ion species (H/He/Be/N etc.)



Wirth, et al. MRS Bulletin 36 (2011) 216-222

## Beryllium First Wall



Tungsten Divertor

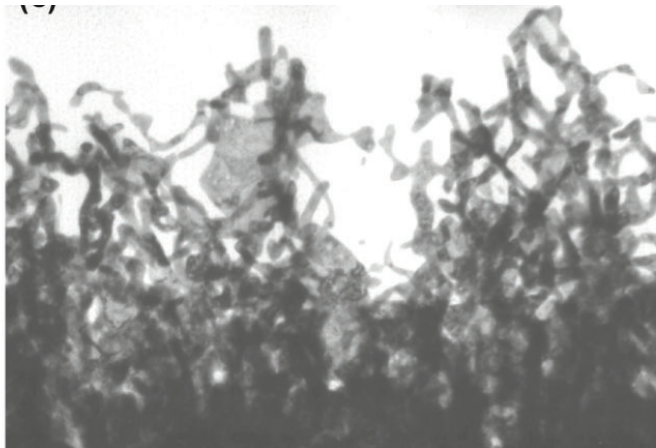
iter.org

- Many complex processes that occur at the plasma/material interface that can lead to material degradation

# Plasma Material Interactions in Tungsten

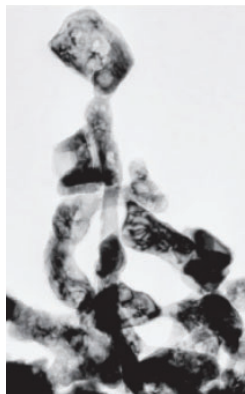
3

## Helium Fuzz Growth



Kajita, et al. J. Nucl. Mater, 418, (2011) 152-158

## Material Degredation

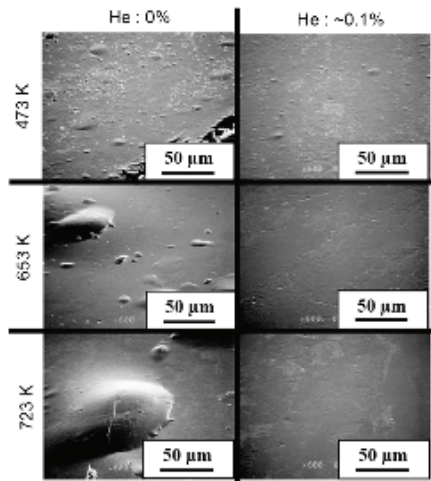


Kajita, et al. Nucl. Fus. 471, 886-890 (2007)

## Tritium Retention



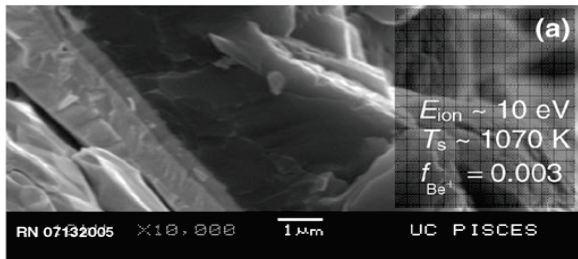
### Effect of He on H Blistering



Ueda, et. al. J. Nucl. Mater. 386-388 (2009) 725-728

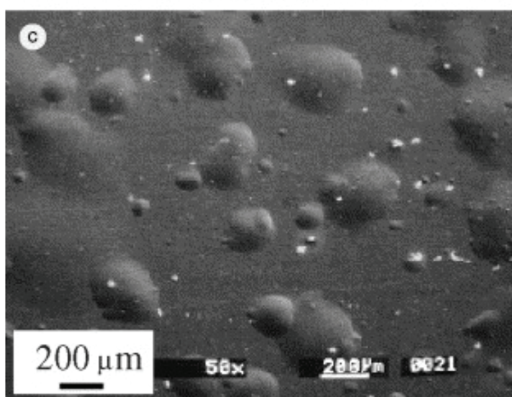
## W-Be Intermetallics

Be<sub>12</sub>W Be deposits (surface)



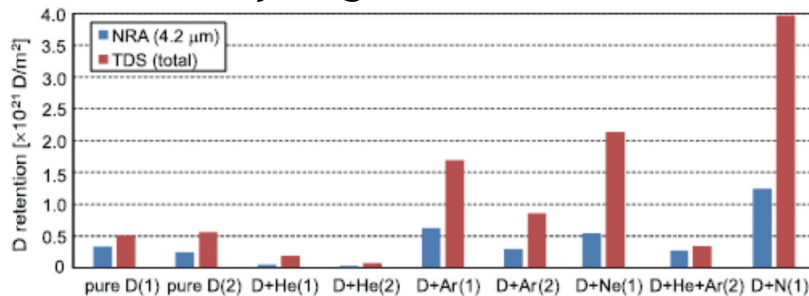
Baldwin, et. al. J. Nucl. Mater. 363-365 (2007) 1179-1183

## Hydrogen Blisters



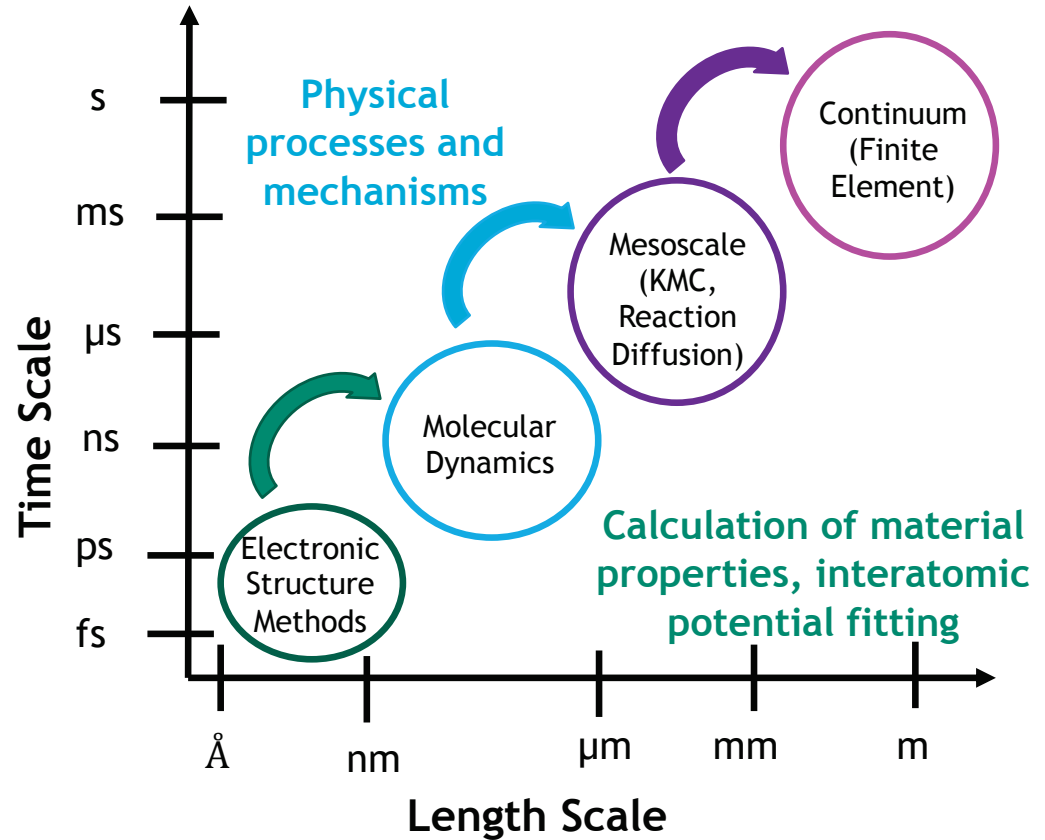
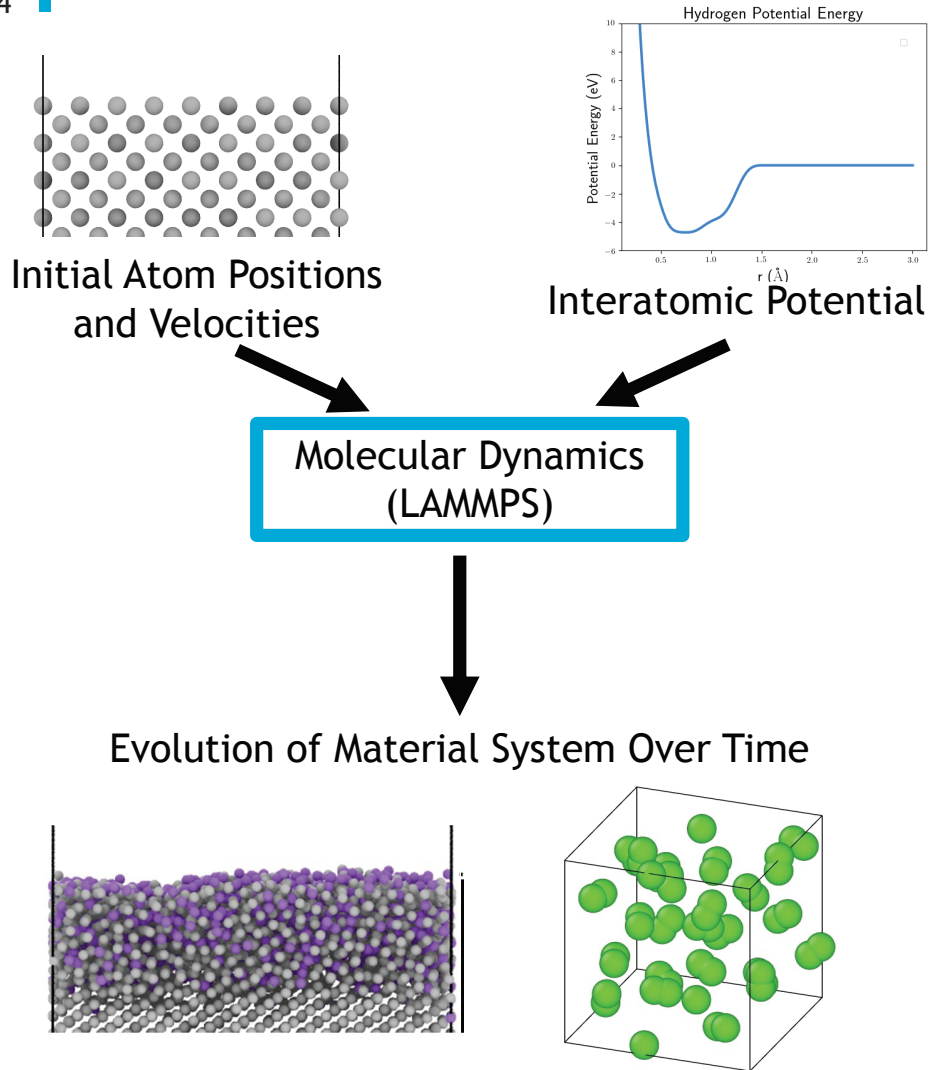
Ye, et al. J. Nucl. Mater. 313-316, 72-76 (2003)

## Effect of Plasma Impurities on Hydrogen Retention



Kreter, et al. Nucl. Fus.. 59, 086029 (2019)

# Multiscale Modeling of Materials

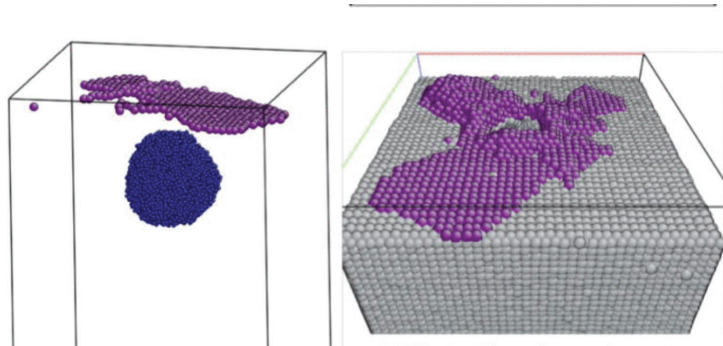


Each simulation technique can provide information to the next scale up

# What Can MD Tell Us About Plasma Material Interactions?

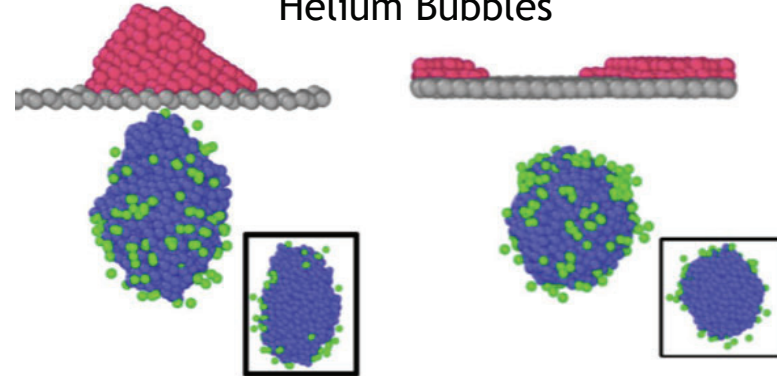


## Helium Bubbles Deform Tungsten Surface



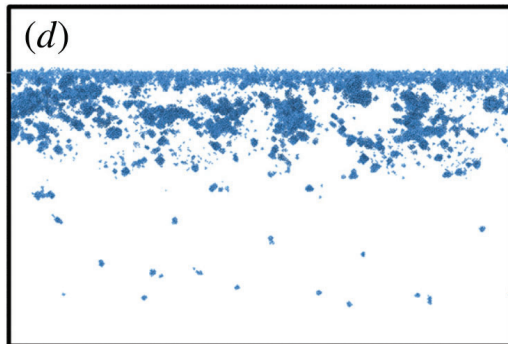
Sefta et al 2013 J. Appl. Phys 114 243518

## Hydrogen Trapping at Helium Bubbles



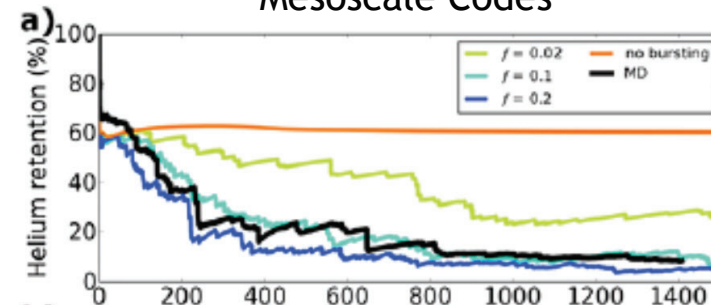
Bergstrom, et al. Fus. Sci. Tech., 71, 122-135, (2017)

## Large-scale simulations of He implantation in W



Hammond et al 2019 Nucl. Fusion 59 066035

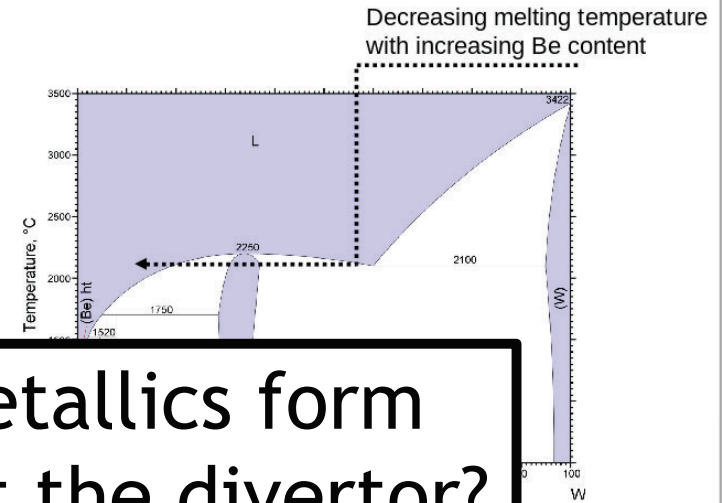
## Using MD to Benchmark Mesoscale Codes



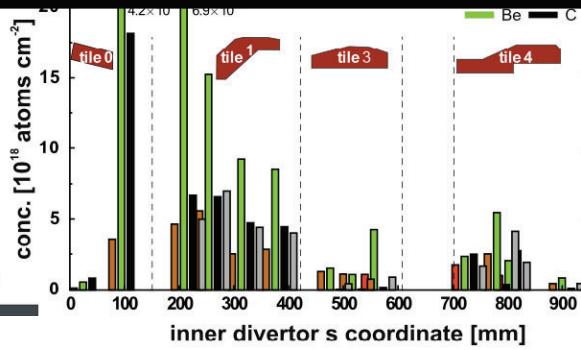
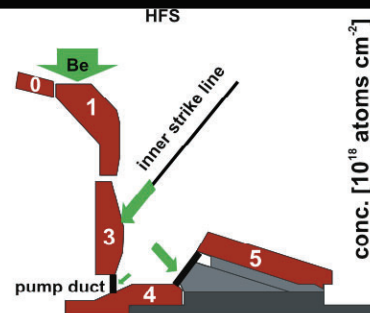
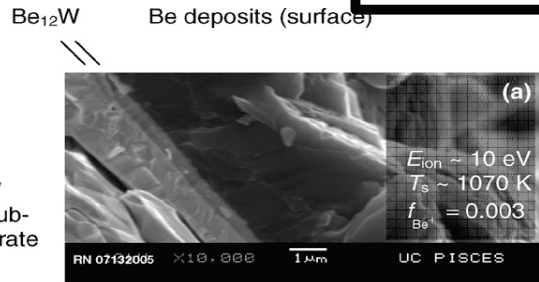
Blondel et al 2018 Nucl. Fus. 58 126034

# 6 Beryllium Effect on Tungsten Melting Temperature

- W-Be intermetallics observed in linear plasma experiments
- Tokamak experiments indicated beryllium deposition on the divertor
- Phase diagram indicates decreasing melting temperature with increasing Be content



**How do W-Be Intermetallics form and how do they affect the divertor?**



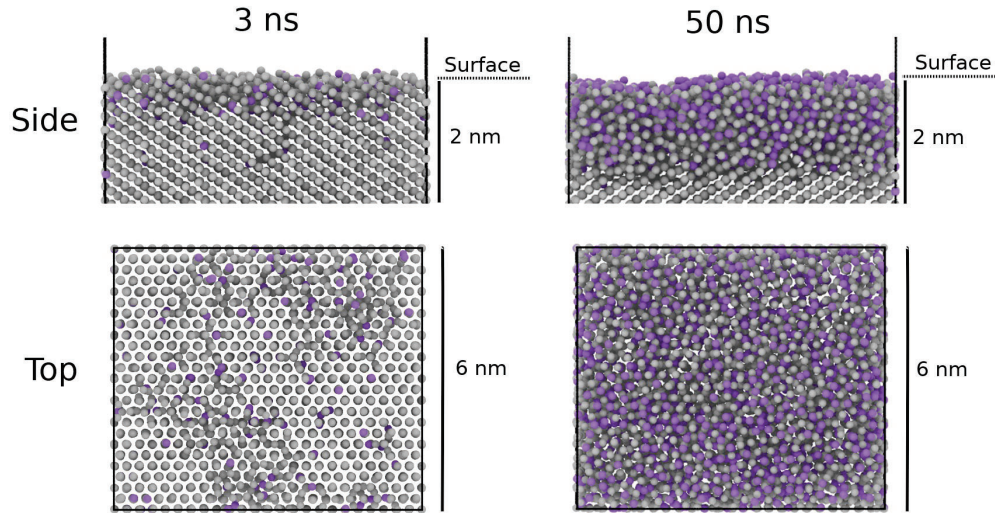
	MD Melt T. (C)	Expt. Melt T (C)
W	3130	3422
Be	1630	1289
WBe <sub>2</sub>	1830	2250

Baldwin, et. al. J. Nucl. Mater. 363-365 (2007) 1179-1183

Brezinsek, et. al. J. Nucl. Mater. 463 (2015) 11-21

# Beryllium Deposition on Tungsten Surfaces

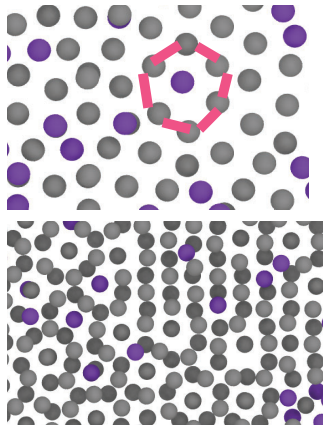
Purple: Be  
Gray: W



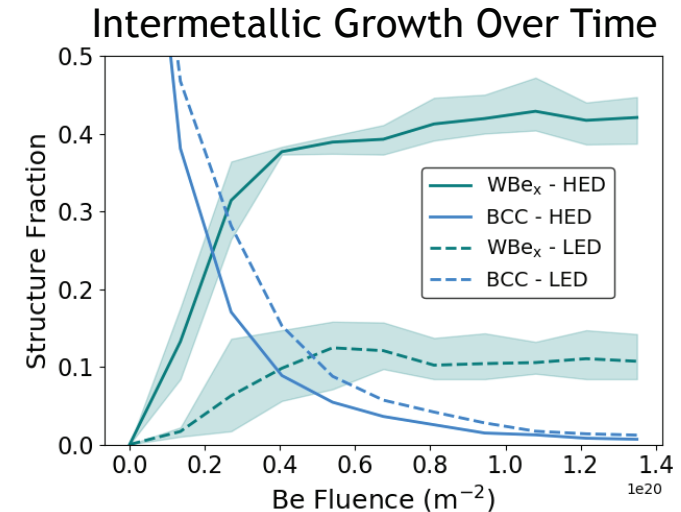
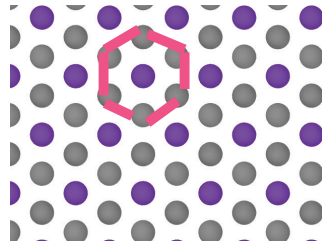
## 75 eV Beryllium Deposition

- Be implantation (75 eV) creates near-surface disordered layer
- No Be diffusion beyond 2 nm
- Ordered structures emerge within disordered layer
- Clear restructuring of near-surface region occurs within nanoseconds

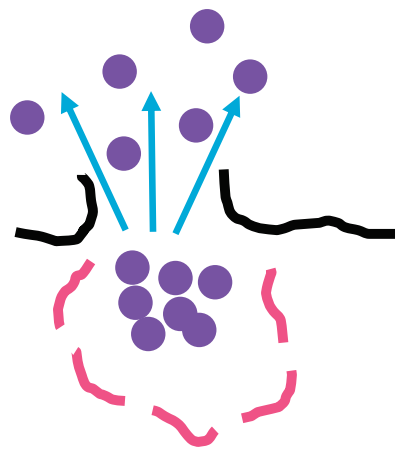
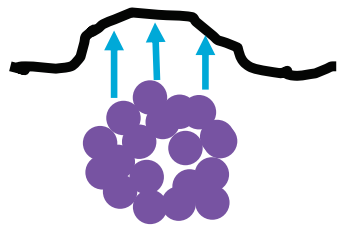
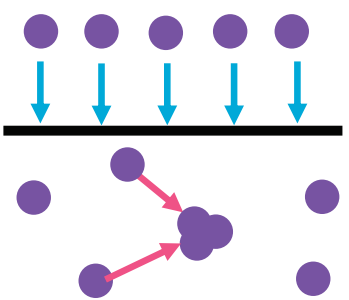
### Implantation Simulations



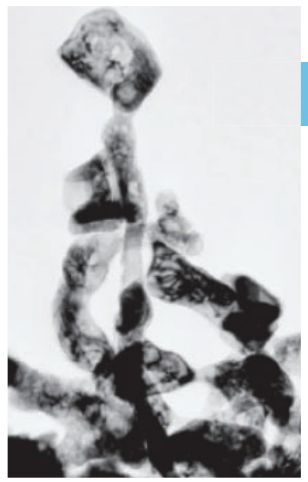
### Expected Ordered Intermetallic Structure



# Mixed Material Effects in the Divertor



Purple: He Atom



Kajita, et al. Nucl. Fus. 471, 886-890 (2007)

$D_2 - 0.1 \text{ He Plasma}$

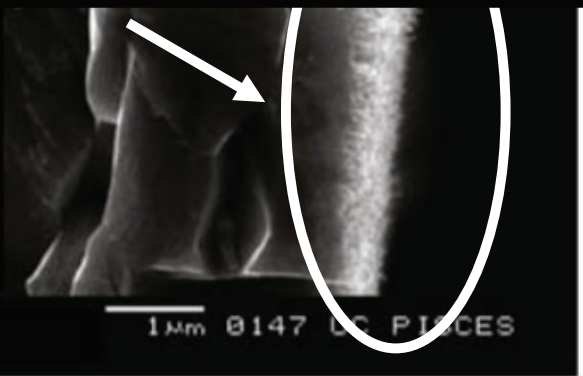
$D_2 - 0.1 \text{ He with Be Plasma}$

W bulk

W bulk

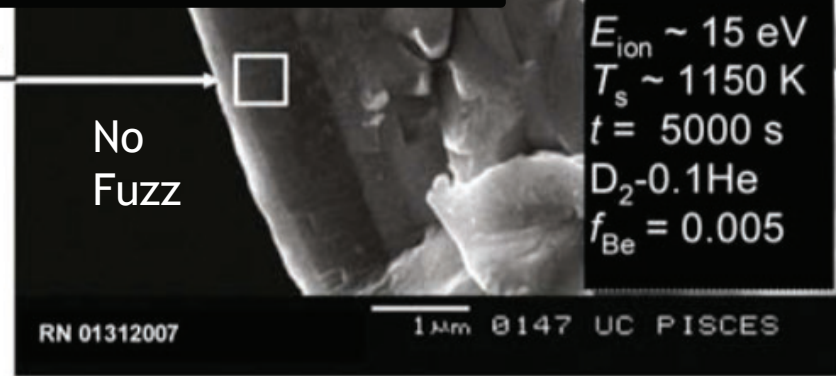
How does beryllium disrupt the helium fuzz process?

$E_{ion} \sim 60 \text{ eV}$   
 $T_s \sim 1150 \text{ K}$   
 $t = 4200 \text{ s}$   
 $D_2-0.1\text{He}$



$Be_{93}W_7$

No Fuzz



$E_{ion} \sim 15 \text{ eV}$   
 $T_s \sim 1150 \text{ K}$   
 $t = 5000 \text{ s}$   
 $D_2-0.1\text{He}$   
 $f_{Be} = 0.005$

RN 02022007

1mm 8147 UC PISCES

RN 01312007

1mm 8147 UC PISCES

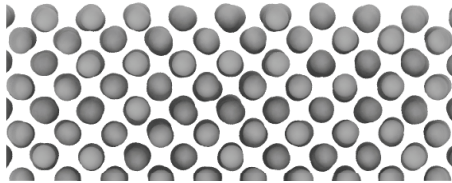


# Cumulative He Implantation in W and W-Be at $2.5 \times 10^{19} \text{ m}^{-2}$

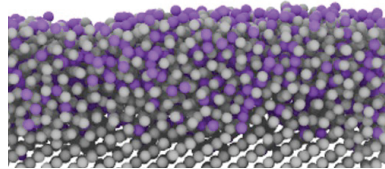


Increasing Time

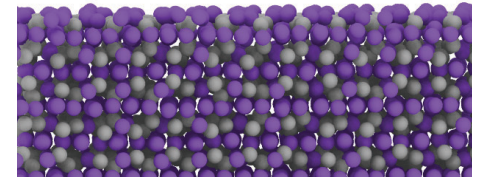
Crystalline W



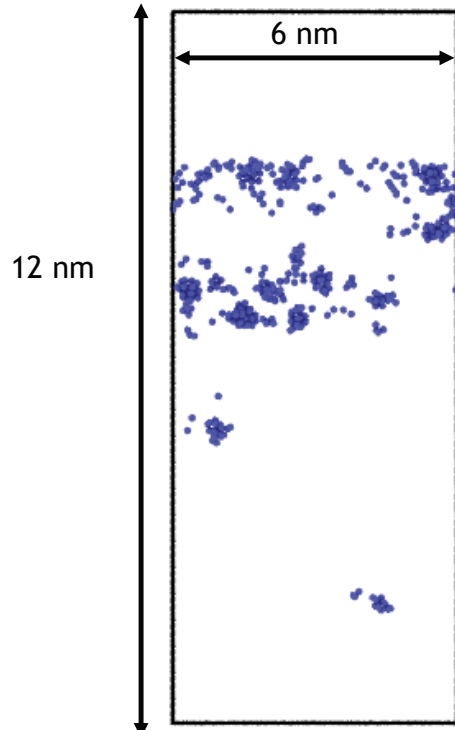
Amorphous W-Be



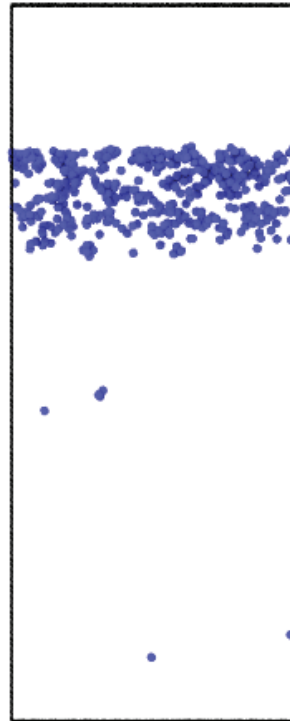
WBe<sub>2</sub> C14 Structure



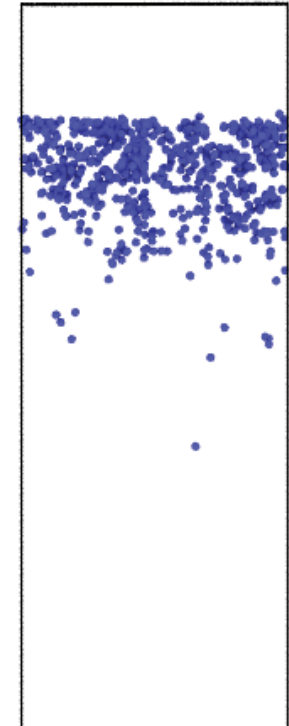
Blue: He  
Purple: Be  
Gray: W



Tungsten:  
Larger He clusters  
distributed  
throughout  
simulation cell



Laves/Deposited  
Layer:  
Smaller He  
clusters mostly  
located near the  
surface



Cusentino, Wood, and  
Thompson, Nucl. Fusion, 60,  
126018 (2020)

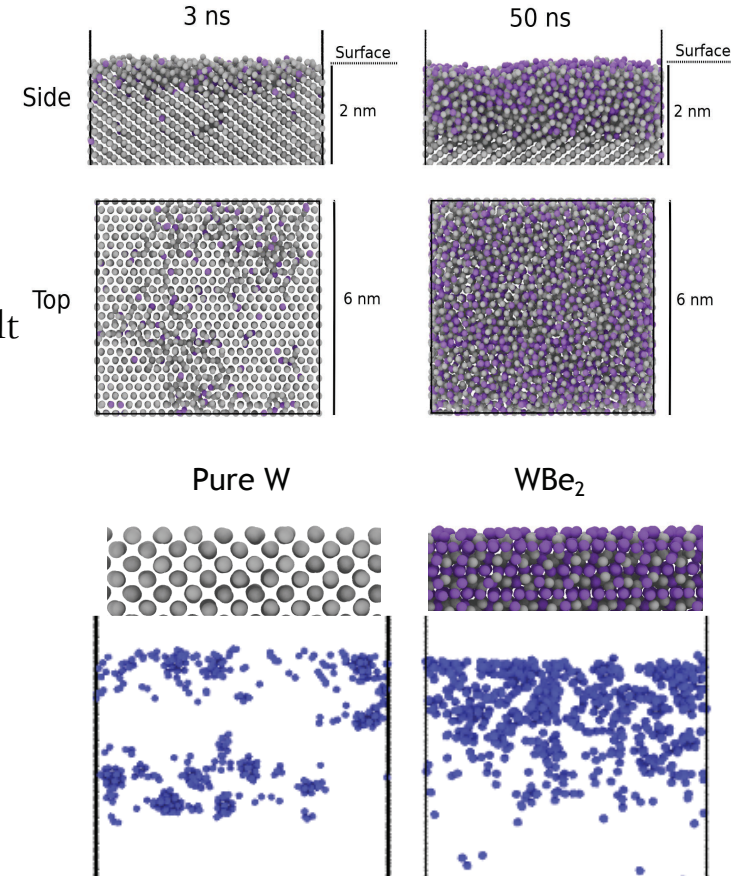
# Summary

- Fusion reactor components are subject to high temperatures and high fluxes of multiple plasma species
- Understanding material degradation and tritium retention in these components is critical for designing fusion reactor components
- Atomistic modeling is a powerful tool that can provide insight into small/short scale processes that drive material degradation and are difficult to observe in experiments
- Simulations of beryllium deposition in tungsten indicated the formation of a disordered layer comparable to experimental observations
- The mixed W-Be layer affected the initial helium diffusion and bubble nucleation and growth which may affect helium fuzz growth

Contact:  
mcusent@sandia.gov



**SciDAC4-PSI2**



# What Makes a Machine Learned Interatomic Potential?



## Training Data

- Generated using quantum methods
- Can include:
  - Energies
  - Forces
  - Stresses
- Variety of atomic configurations
  - Bulk structures, liquids, surfaces, defects, etc.

## Descriptor

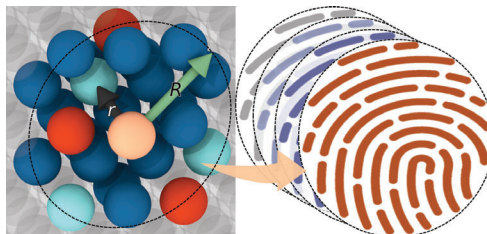
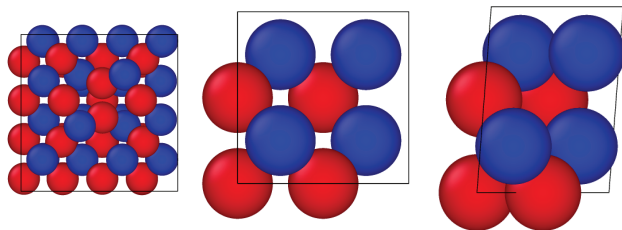
- Describes the local atomic environment
- Requirements
  - Rotation/Translation/Permutation invariant
  - Equivariant forces
  - Smooth differentiable
  - Extensible
- Some Examples
  - Bispectrum, SOAP, ACE, Moment Tensors, etc.

## Regression Method

- Linear regression
- Kernel ridge regression
- Gaussian process
- Non-linear optimization
- Neural Networks

## SNAP

- Energies, forces, and stresses from DFT
- Bispectrum component descriptors
- Linear regression



# SNAP Definition and Work Flow

## Model Form

- Energy of atom  $i$  expressed as a basis expansion over  $K$  components of the bispectrum ( $B_k^i$ )

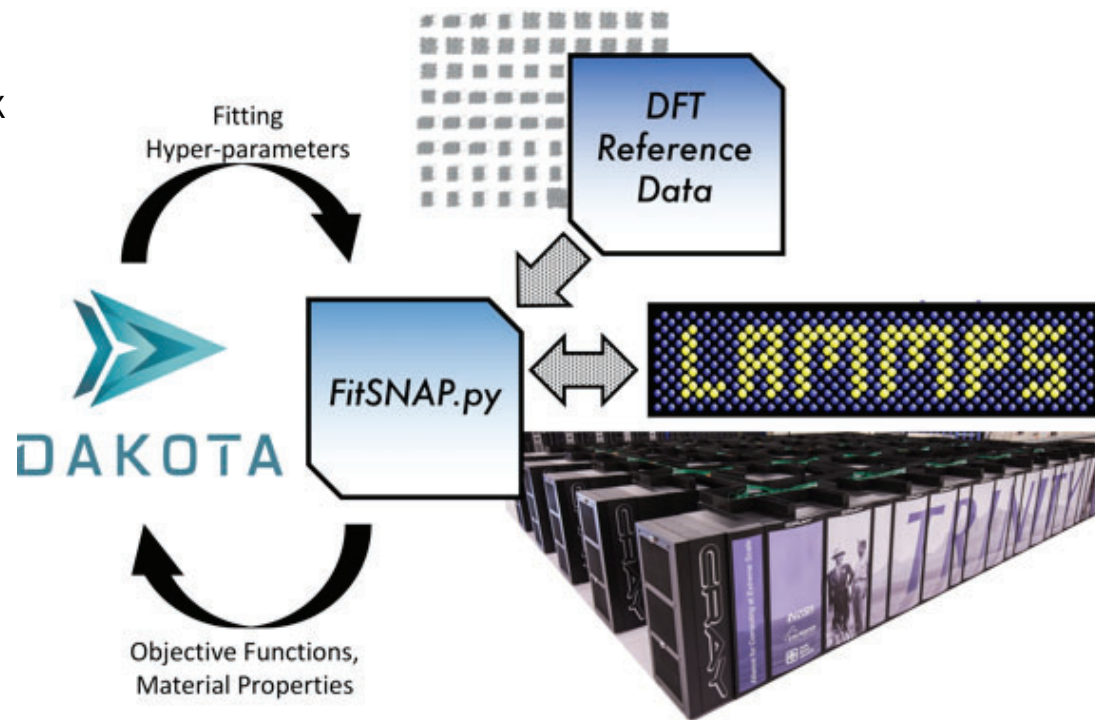
$$E_{SNAP}^i = \beta_0 + \sum_{k=1}^K \beta_k (B_k^i - B_{k0}^i)$$

## Regression Method

- $\beta$  vector fully describes a SNAP potential
- Decouples MD speed from training set size

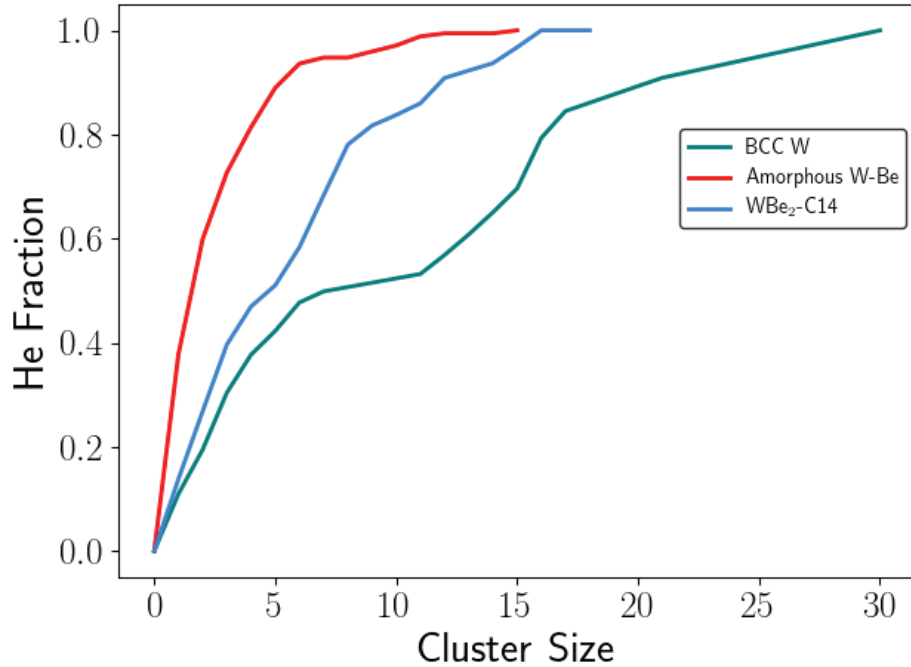
$$\min(\|\mathbf{w} \cdot D\boldsymbol{\beta} - T\|^2 - \gamma_n \|\boldsymbol{\beta}\|^n)$$

Weights    Set of Descriptors    DFT Training



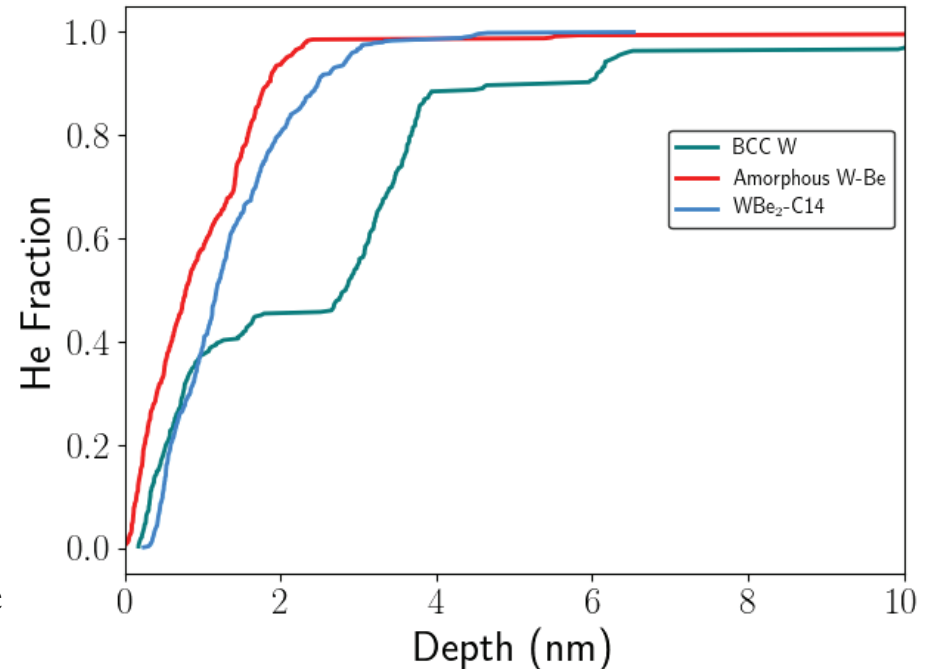
Code available: <https://github.com/FitSNAP/FitSNAP>

# Cumulative He Implantation in W and W-Be at $2.5 \times 10^{19} \text{ m}^{-2}$



- Larger He clusters in pure tungsten and more, smaller clusters when Be is present
- Retention is much higher when Be is present
- He mostly remains within 2 nm of surface when Be is present

	He % Retention	Number of Clusters
W	36.7	85
Amorphous W-Be	49.3	172
WBe <sub>2</sub> - C14	67.3	182



# He Properties Change in W-Be Intermetallics

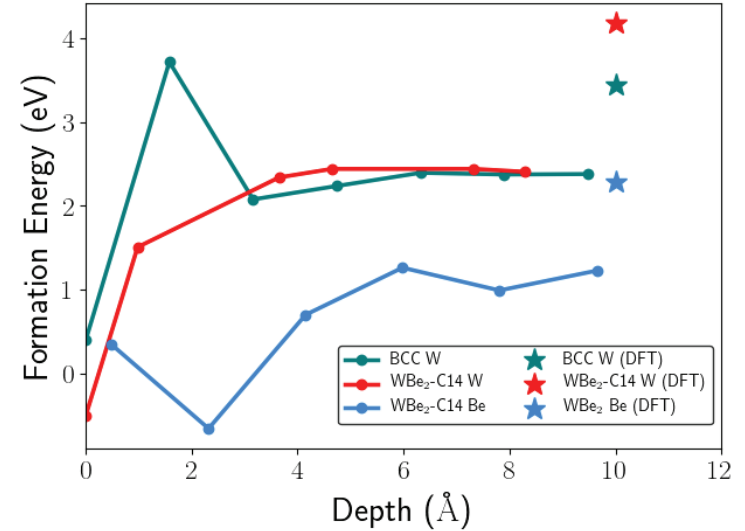
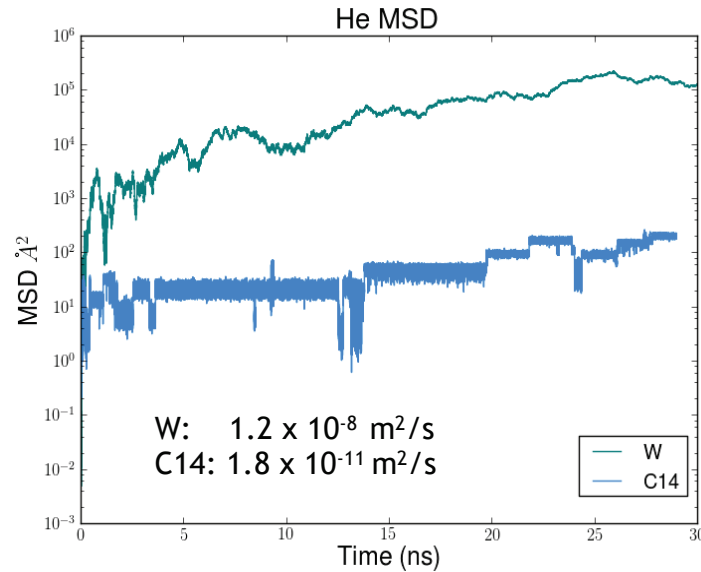
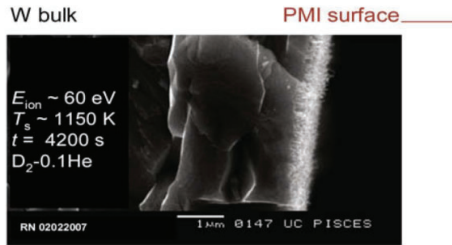


Lack of fuzz growth when Be is in plasma

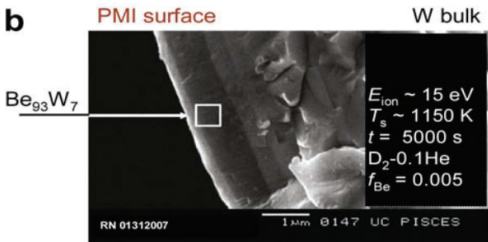
He diffusion is slower in  $WBe_2$

Vacancy formation energy is lower in  $WBe_2$

$D_2$  - 0.1 He Plasma



$D_2$  - 0.1 He with Be Plasma



Baldwin, et al. J. Nucl. Mater 390-391, 886-890 (2009)

Combination of slower diffusion and lower vacancy formation energy in W-Be structures leads to smaller, more numerous He bubbles that remain near the surface